$\cdots \mathrm{H}(\mathrm{C} 20)$ of $2.05 \AA$ would be even less if the angle $\mathrm{C}(14)-\mathrm{C}(18)-\mathrm{C}(20)$ had the expected value of $115^{\circ}$ rather than the observed $118^{\circ}$.

The most dominant feature in the packing is the hydrogen-bond formation $\mathrm{O}-\mathrm{H} \cdots \mathrm{N}$ which is essentially the same as in the $d l$-compound, although the six-membered ring, which is planar within $0 \cdot 15 \AA$, has lost its centre of symmetry (Fig. 4). The distances $\mathrm{O}(1) \cdots \mathrm{N}(2)$ and $\mathrm{O}(2) \cdots \mathrm{N}(1)$ are 2.793 (5) and 2.806 (6) $\AA$ respectively.

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# Compounds with Perovskite-Type Slabs. II. The Crystal Structure of $\mathbf{S r}_{2} \mathbf{T a}_{2} \mathbf{O}_{7}$ 

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#### Abstract

The crystal structure of $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ has been determined from three-dimensional X-ray diffraction data and refined to a conventional $R$ value of 0.036 for 697 observed reflexions. $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ crystallizes in the orthorhombic space group $C m c m$, with $a=3 \cdot 937$ (6), $b=27 \cdot 198$ (6), $c=5 \cdot 692$ (7) $\AA$ and $Z=4$. The crystal is essentially isostructural with $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$, having a structure composed of perovskite-type slabs parallel to ( 010 ). There are two independent Ta atoms surrounded octahedrally by O atoms. The $\mathrm{Ta}(1)-\mathrm{O}$ and $\mathrm{Ta}(2)-\mathrm{O}$ distances range from 1.87 (1) to $2 \cdot 16$ (1) $\AA$ and 1.89 (1) to 2.07 (1) $\AA$ respectively. The $\operatorname{Sr}(2)$ atom is surrounded by 12 O atoms with distances ranging from 2.70 (3) to 2.85 (1) $\AA$. The $\mathrm{Sr}(1)$ atom near the boundary of a slab is surrounded by six O atoms in the same slab and two O atoms in the neighbouring slab with distances ranging from $2 \cdot 47$ (1) to $3 \cdot 27$ (1) $\AA$.


## Introduction

Recently, some $\mathrm{A}_{2} \mathrm{~B}_{2} \mathrm{O}_{7}$ compounds were revealed to form a structural family characterized by the perov-skite-type slabs in their structures. Examples are $\mathrm{Ca}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ (Scheunemann \& Müller-Buschbaum, 1974), $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ (Ishizawa, Marumo, Kawamura \& Kimura, 1975) and $\mathrm{La}_{2} \mathrm{Ti}_{2} \mathrm{O}_{7}$ (Gasperin, 1975). Though crystals of $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ had been supposed to have a tetragonal pyrochlore-type structure with $a=10 \cdot 63, c=$ 10.91 A (Ismailzade, 1958), it was found by Nanamatsu, Kimura \& Kawamura (1975) that the symmetry and the cell dimensions rather suggest a close similarity in the structure to $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7} . \mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ is paraelectric at room temperature, having a markedly lower ferroelectric phase transition temperature $\left(-107^{\circ} \mathrm{C}\right)$ than $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7} \quad\left(1342^{\circ} \mathrm{C}\right), \quad \mathrm{La}_{2} \mathrm{Ti}_{2} \mathrm{O}_{7}$ $\left(\sim 1500^{\circ} \mathrm{C}\right)$ and $\mathrm{Ca}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}\left(>1500^{\circ} \mathrm{C}\right)$ (Nanamatsu et al., 1975). Therefore, the crystal structure of $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ at room temperature is expected to afford important information on the structures of the paraelectric phases of these compounds.

## Experimental

Transparent, colourless crystals were synthesized from a $2: 1$ mixture of $\mathrm{SrCO}_{3}$ and $\mathrm{Ta}_{2} \mathrm{O}_{5}$ with a floatingzone technique. From Weissenberg photographs, the crystal was found to have orthorhombic symmetry, and systematic absences, $h+k$ odd in $h k l$ and $l$ odd in $h 0 l$, restricted the possible space groups to $C 2 \mathrm{~cm}$ (No. 40), $C m c 2_{1}$ and $C m c m$. Since the crystal is paraelectric at room temperature, the space group Cmcm seemed to be most plausible, and has been justified in the course of the structure determination. The cell dimensions were determined from $152 \theta$ values in the range between 20 and $70^{\circ}$ measured with $\mathrm{Cu} K \alpha$ radiation on a powder diffractometer, by the least-squares procedure. The crystal data are given in Table 1.

Intensities were collected on a four-circle diffractometer (Philips PW 1100) with Mo $K \alpha$ radiation from a graphite monochromator. A crystal with dimensions of about $0.166 \times 0.024 \times 0.166 \mathrm{~mm}$ was used. The $\omega-2 \theta$ scan technique was employed with a scanning speed of $4^{\circ} \mathrm{min}^{-1}$ in $\omega$. In all, 697 independent reflexion

Table 1. Crystal data of $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$
Crystal system: orthorhombic
Space group: Cmcm
Cell dimensions

$$
\begin{aligned}
a & =3 \cdot 937(6) \AA \\
b & =27 \cdot 198(6) \\
c & =5 \cdot 692(7) \\
U & =609 \cdot 5 \AA^{3}
\end{aligned}
$$

$Z=4$
M.W. $649 \cdot 16$
$D_{x}=7.07 \mathrm{~g} \mathrm{~cm}^{-3}$
$\mu\left(\right.$ for $\mathrm{Mo} \mathrm{K} \mathrm{\alpha}$ ) $=555.4 \mathrm{~cm}^{-1}$
data, whose $|F|$ 's were larger than $3 \sigma(|F|)$, were obtained within the range of $2 \theta<100^{\circ} . \sigma(|F|)$ is the estimated standard deviation of $|F|$ due to counting statistics. Intensities were corrected for the Lorentz and polarization factors. Absorption corrections were also carried out with the program $A C A C A-4$ (Wuensch \& Prewitt, 1965), by taking the crystal shape into account. The maximum and minimum transmission factors for the observed reflexions were 0.2545 and 0.0373 respectively.

## Structure determination

The metal atom positions were derived from Patterson maps. They were concordant with the centrosymmetric space group Cmcm . All the O atoms were found on the electron-density maps synthesized with phases calculated with the heavy atoms. The structure was refined with a full-matrix least-squares program, LINUS (Coppens \& Hamilton, 1970), to a conventional $R$ value of 0.061 , with isotropic temperature factors for all the atoms. At this stage, the possibility of slight deviations from the centrosymmetric atomic arrangements was examined. Refinements based on the space group $C m c 2_{1}$ or $C 2 \mathrm{~cm}$ gave negative or unreasonably large temperature factors for some O atoms. In addition, the $R$ values were even larger for these models than for the centrosymmetric one. Therefore the non-centrosymmetric models were discarded. Further refinements were undertaken introducing anisotropic temperature factors for metal atoms as well as the isotropic secondary exinction parameter, and were terminated when all the calculated shifts became less than one tenth of the estimated standard deviations. The extinction parameter obtained was $0.3 \times 10^{-4}$ and the largest correction factor was about $35 \%$ of $F$.

Scattering factors for $\mathrm{Sr}^{2+}$ and $\mathrm{Ta}^{5+}$ ions were taken from International Tables for X-ray Crystallography (1974) and the values given by Tokonami (1965) were used for $\mathrm{O}^{2-}$. Unit weight was assigned for all the terms. The final positional and thermal parameters are listed in Table 2.* Selected bond lengths and angles are given in Table 3 with their estimated standard deviations in parentheses.

Table 2. Final atomic parameters
Anisotropic temperature factors are expressed in the form: $\exp \left[-\left(h^{2} B_{11}+k^{2} B_{22}+l^{2} B_{33}+2 h k B_{12}+2 h l B_{13}+2 k l B_{23}\right)\right]$.

|  | $x$ | $y$ | $z$ | $B$ |
| :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Sr}(1)$ | $0 \cdot 0$ | $0 \cdot 2884$ (1) | $0 \cdot 25$ | 0.87* |
| Sr(2) | $0 \cdot 0$ | $0 \cdot 4468$ (1) | 0.75 | 0.99* |
| $\mathrm{Ta}(1)$ | $0 \cdot 5$ | $0 \cdot 3399$ (1) | $0 \cdot 75$ | 0.53* |
| $\mathrm{Ta}(2)$ | 0.5 | $0 \cdot 4447$ (1) | $0 \cdot 25$ | 0.42* |
| $\mathrm{O}(1)$ | 0.5 | $0 \cdot 2930$ (6) | 0.510 (2) | 0.9 (1) |
| $\mathrm{O}(2)$ | $0 \cdot 0$ | 0.3475 (10) | 0.75 | $1 \cdot 5$ (3) |
| $\mathrm{O}(3)$ | $0 \cdot 5$ | $0 \cdot 4009$ (6) | $0 \cdot 508$ (2) | $1 \cdot 2$ (2) |
| $\mathrm{O}(4)$ | $0 \cdot 0$ | 0.4524 (10) | $0 \cdot 25$ | $1 \cdot 3$ (2) |
| O(5) | $0 \cdot 5$ | $0 \cdot 5$ | $0 \cdot 5$ | $1 \cdot 1$ (2) |
|  | $B_{11}$ | $B_{22}$ | $B_{33} \quad B_{12}$ | $B_{13}=B_{23}$ |
| $\mathrm{Sr}(1)$ | $0 \cdot 0077$ (6) | $0 \cdot 00058$ (6) | 0.0033 (3) | $0 \cdot 0$ |
| $\mathrm{Sr}(2)$ | 0.0062 (6) | 0.00053 (6) | 0.0079 (4) | 0.0 |
| $\mathrm{Ta}(1)$ | $0 \cdot 0037$ (2) | 0.00036 (3) | $0 \cdot 0022$ (1) | $0 \cdot 0$ |
| $\mathrm{Ta}(2)$ | $0 \cdot 0027$ (2) | $0 \cdot 00031$ (2) | 0.0013 (1) | $0 \cdot 0$ |

* Calculated from anisotropic temperature factors according to the formula: $B=4\left(B_{11} a^{2}+B_{22} b^{2}+B_{33} c^{2}\right) / 3$.


## Discussion

The projections of the structure along the $a$ and $c$ axes are shown in Fig. 1 (a) and (b) respectively. The projection of the structure of $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ along the $a$ axis is also given in Fig. 1(c) for comparison. The structure is composed of perovskite-type slabs with a thickness corresponding to twice the face diagonal of a perovskite cube. The slabs consist of $\mathrm{TaO}_{6}$ octahedra connected by sharing corners, with $\mathrm{Sr}(2)$ atoms in the spaces of the twelvefold coordination. $\mathrm{The} \mathrm{Sr}(1)$ atoms are near the boundaries of the slabs.

* A list of structure factors has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 31741 ( 6 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 13 White Friars, Chester CH1 1NZ, England.

Table 3. Interatomic distances $(\AA)$ and bond angles $\left({ }^{\circ}\right)$ with their estimated standard deviations in parentheses

| $\mathrm{Ta}(1)-\mathrm{O}(1)$ | 1.87 (1) | $(\times 2)$ | $\mathrm{Sr}(2)-\mathrm{O}(2)$ | 2.70 (3) |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ta}(1)-\mathrm{O}(2)$ | 1.979 (4) | $(\times 2)$ | $\mathrm{Sr}(2)-\mathrm{O}\left(4^{\text {III }}\right.$ ) | 2.74 (3) |  |
| $\mathrm{Ta}(1)-\mathrm{O}(3)$ | $2 \cdot 16$ (1) | $(\times 2)$ | $\mathrm{O}(2)-\mathrm{Ta}(1)-\mathrm{O}(1)$ | $94 \cdot 1$ (5) | $(\times 4)$ |
| $\mathrm{Sr}(1)-\mathrm{O}(1)$ | $2 \cdot 47$ (1) | $(\times 4)$ | $\mathrm{O}(2)-\mathrm{Ta}(1)-\mathrm{O}(3)$ | 85.4 (6) | $(\times 4)$ |
| $\mathrm{Sr}(1)-\mathrm{O}\left(1^{\text {iv }}\right.$ ) | $2 \cdot 60$ (1) | $(\times 2)$ | $\mathrm{O}(1)-\mathrm{Ta}(1)-\mathrm{O}(3)$ | $93 \cdot 4$ (5) | $(\times 2)$ |
| $\mathrm{Sr}(1)-\mathrm{O}(2)$ | $3 \cdot 27$ (1) | $(\times 2)$ | $\mathrm{O}(1)-\mathrm{Ta}(1)-\mathrm{O}\left(1^{1}\right)$ | 94.0 (6) |  |
| $\mathrm{Ta}(2)-\mathrm{O}(3)$ | $1 \cdot 89$ (1) | $(\times 2)$ | $\mathrm{O}(3)-\mathrm{Ta}(1)-\mathrm{O}\left(3^{1}\right)$ | $79 \cdot 3$ (5) |  |
| $\mathrm{Ta}(2)-\mathrm{O}(4)$ | 1.980 (4) | $(\times 2)$ | $\mathrm{O}(4)-\mathrm{Ta}(2)-\mathrm{O}(3)$ | $93 \cdot 8$ (5) | ( $\times 4$ ) |
| $\mathrm{Ta}(2)-\mathrm{O}(5)$ | 2.070 (1) | $(\times 2)$ | $\mathrm{O}(4)-\mathrm{Ta}(2)-\mathrm{O}(5)$ | $85 \cdot 6$ (5) | ( $\times 4$ ) |
| $\mathrm{Sr}(2)-\mathrm{O}(3)$ | 2.71 (1) | $(\times 4)$ | $\mathrm{O}(3)-\mathrm{Ta}(2)-\mathrm{O}(5)$ | $85 \cdot 6$ (4) | $(\times 2)$ |
| $\mathrm{Sr}(2)-\mathrm{O}(5)$ | 2.827 (3) | $(\times 4)$ | $\mathrm{O}(3)-\mathrm{Ta}(2)-\mathrm{O}\left(3^{\text {i }}\right.$ ) | 102.0 (6) |  |
| $\mathrm{Sr}(2)-\mathrm{O}(4)$ | $2 \cdot 850$ (4) | $(\times 2)$ | $\mathrm{O}(5)-\mathrm{Ta}(2)-\mathrm{O}\left(5^{\text {i }}\right.$ ) | $86 \cdot 8$ (1) |  |

Symmetry code: None $x, y, z$; (i) $0 \cdot 5, y, 1 \cdot 5-z$; (ii) $0 \cdot 5, y,-0 \cdot 5+z$; (iii) $0,1-y, 1-z$; (iv) $0,0 \cdot 5-y, 1-z$.

Adjacent slabs are mutually shifted by $a / 2$. All the atoms of $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ are on the mirror planes at $x=0$ and $0 \cdot 5$. All these features are the same as in the structure of $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ (Ishizawa et al., 1975). In $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ the metal atoms and $O(2)$ and $O(4)$ are also on the mirror planes at $z=0.25$ and 0.75 . $\mathrm{O}(5)$ is at the centre of symmetry. Only $O(1)$ and $O(3)$ have $z$ parameters, although their actual shifts from the plane $z=0.5$ are very small, i.e. $0.06(1)$ and $0.05(1) \AA$ respectively. The $\mathrm{Sr}(2)-\mathrm{O}$ distances are in the range $2 \cdot 70(3)-2 \cdot 850(4) \AA$. The $\mathrm{Sr}(1)$ atom is surrounded by six O atoms in the same slab and two in the neighbouring slab with distances between $2 \cdot 47(1)$ and $3 \cdot 27(1) \AA$.
$\mathrm{Ta}(1)$ and $\mathrm{Ta}(2)$ atoms are displaced from the centre of respective $\mathrm{TaO}_{6}$ octahedra toward the boundary of the slab by about $0 \cdot 20$ and $0 \cdot 17 \AA$ respectively (Fig. $2 a$ ). These displacements of Ta atoms result in the formation of two short and two long bonds on the mirror planes in each octahedron. In the case of $\mathrm{Ta}(1) \mathrm{O}_{6}$ they are $1.87(1) \quad[\mathrm{Ta}(1)-\mathrm{O}(1), \times 2]$, and $2 \cdot 16(1) \AA[\mathrm{Ta}(1)-\mathrm{O}(3), \times 2]$, and in $\mathrm{Ta}(2) \mathrm{O}_{6}, 1 \cdot 89(1)$ $[\mathrm{Ta}(2)-\mathrm{O}(3), \times 2]$ and $2 \cdot 070(1) \AA[\mathrm{Ta}(2)-\mathrm{O}(5), \times 2]$ respectively. The two remaining $\mathrm{Ta}-\mathrm{O}$ distances are nearly equal to the mean bond length in each octahedron. $\operatorname{Sr}(2)$ is also displaced from the centre of the coordination polyhedron toward the boundary of the slab by $0.09 \AA$. These cation displacements are supposed to be chiefly due to the repulsive forces between the $\mathrm{Ta}^{5+}$ and $\mathrm{Sr}^{2+}$ cations.

The most conspicuous structural difference between $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ and $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ is in the degree of deformation of the perovskite-type layers. In $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$, the arrangement of O atoms is largely deformed from that in the ideal perovskite structure, in contrast to the nearly regular arrangement in $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$, as seen in Fig. 1. The displacements of Nb atoms from the centres of $\mathrm{NbO}_{6}$ octahedra have $z$ components as well as $y$ components (Fig. 2b), and consequently the mirror planes at $z=0.25$ and 0.75 observed in $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ are lost in $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$. The centrosymmetric structure obtained explains well the paraelectric character of the $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ crystals at room temperature.

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Fig. 1. The crystal structure of $\mathrm{Sr}_{2} \mathrm{Ta}_{2} \mathrm{O}_{7}$ viewed (a) along $a$, (b) along $c$, and (c) the structure of $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ viewed along $a$. The unit cell of $\mathrm{Sr}_{2} \mathrm{Nb}_{2} \mathrm{O}_{7}$ is moved along the $c$ axis for comparison, since the origin can be taken at any point on the twofold screw axis. The thick circles in $(a)$ and $(c)$ are at the level $x=0$, and the others at $x=0 \cdot 5$. Atoms are labelled in accordance with the nomenclature of Table 2.


Fig. 2. (a) The displacements of the Ta atoms and (b) those of the Nb atoms from the centres of respective octahedra in a slab.

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